

# TPD and TPR Studies of Molybdenum Nitride

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The surface species of passivated and nonpassivated molybdenum nitride samples and their stability at elevated temperatures have been investigated by temperature-programmed decomposition and temperature-programmed reduction with H<sub>2</sub>. Large differences between fresh and passivated Mo<sub>2</sub>N were found. (i) Two types of NH<sub>x</sub> adsorbed species remain on the freshly prepared Mo<sub>2</sub>N. The first type, weakly adsorbed, desorbed at 200–300°C. Another type is strongly bonded NH<sub>x</sub> on the surface of Mo<sub>2</sub>N, which decomposed to H<sub>2</sub> and N<sub>2</sub> at 560°C. For passivated Mo<sub>2</sub>N samples, only weakly adsorbed NH<sub>x</sub> species are present. (ii) An obvious change in the crystalline phase of fresh Mo<sub>2</sub>N occurred when it was heated at 850°C in He, while for the passivated Mo<sub>2</sub>N sample no change was observed. When Mo<sub>2</sub>N was heated to 1000°C, the resulting solid consisted of β-Mo<sub>2</sub>N and metallic Mo. Mo<sub>2</sub>N is completely reduced to metallic Mo at 880°C in H<sub>2</sub>. (iii) The H<sub>2</sub> consumption peak around 470°C for passivated Mo<sub>2</sub>N samples is attributed to the reduction of the passivating oxygen. It is suggested that the main effect of passivation on fresh Mo nitride is to eliminate most of the NH<sub>x</sub> and H species strongly bonded on nitride and then forming a protective oxide layer. The oxygen incorporated during passivation can be removed by H<sub>2</sub> at a moderate temperature. © 1997 Academic Press

## 1. INTRODUCTION

Molybdenum nitride as a catalytic material of potential significance has recently received a great deal of attention. It was reported that high surface area molybdenum nitride exhibited outstanding performances for various reactions including NH<sub>3</sub> synthesis (1–3), CO hydrogenation (4), ethane hydrogenolysis (5, 6), quinoline hydrodenitrogenation (7–10), and thiophene hydrodesulfurization (11, 12). While most of the studies focus on the preparation and catalytic activity, little attention has been paid to the influence of passivation treatment upon the surface species existing on molybdenum nitride and its stability at elevated temperature. As fresh molybdenum nitride is very sensitive to oxygen, and rapid autothermal reaction with oxygen leads to serious sintering, in many cases Mo<sub>2</sub>N is passivated before use. Therefore, the surfaces of stable molybdenum nitride have been already altered by oxygen during passivation and some changes have occurred.

It was reported that a protective layer of oxide would form on the surface of Mo<sub>2</sub>N which has undergone passivation treatment. The amount of oxygen present in passivated Mo<sub>2</sub>N was even estimated as large as 19% of the total amount of nitrogen present in pure Mo<sub>2</sub>N (11). But the influences on the surface species of Mo<sub>2</sub>N caused by passivation are not fully understood. Furthermore, before passivated Mo<sub>2</sub>N is employed as a catalyst in catalysts reactions, it is usually activated by H<sub>2</sub> flowing at elevated temperatures, such H<sub>2</sub> reduction at 400°C. It is also not clear whether oxygen incorporated during passivation can be removed by H<sub>2</sub>, and what changes of surface species for passivated Mo<sub>2</sub>N during treatment in H<sub>2</sub> took place. Therefore, it is useful to study the nature of the surface species formed on molybdenum nitride during its preparation, the changes caused by passivation, and the effect of pretreatment on passivated Mo<sub>2</sub>N as well.

The synthesis methodology of molybdenum nitride, using temperature-programmed reaction of MoO<sub>3</sub> with NH<sub>3</sub>, reported by Volpe and Boudart (13), has been widely adopted. Temperature-programmed decomposition (TPD) and temperature-programmed surface reaction (TPR) with H<sub>2</sub> are useful techniques for determining the surface species and their reactivity on molybdenum nitride. In this paper, we report the results of TPD and TPR obtained on fresh and passivated molybdenum nitride samples, and discuss the effect of the passivating process on the surface species of molybdenum nitride and the changes in the composition at elevated temperatures with the aid of XRD analysis.

## 2. EXPERIMENTAL

### 2.1. Preparation of Molybdenum Nitride

Molybdenum nitride was prepared by temperature-programmed reaction of MoO<sub>3</sub> with NH<sub>3</sub>. As the nitration of MoO<sub>3</sub> with NH<sub>3</sub> occurs in two steps (14), a two-segment heating program was used to synthesize the molybdenum nitride. After a first quick increase (5°C/min) of the reaction temperature from room temperature to 350°C, the heating rate was controlled at 0.5–1.0°C/min from 350 to 500°C, and 3°C/min from 500 to 700°C. The temperature was kept at 700°C for 2–12 h.

TABLE 1

BET Surface Area and Average Pore Diameter of Mo<sub>2</sub>N

Samples <sup>a</sup>	Heating rate (350–500°C) (°C/min)	NH <sub>3</sub> rate (mol NH <sub>3</sub> /mol MoO <sub>3</sub> /h)	Surface area (m <sup>2</sup> /g)	Average pore diameter (nm)
Mo <sub>2</sub> N-A1	1.0	15	22.6	8.9
Mo <sub>2</sub> N-A2	0.5	22	76.9	4.9
Mo <sub>2</sub> N-A3	0.5	46	103.5	4.8
Mo <sub>2</sub> N-B	0.6		133.3	3.8

<sup>a</sup> The symbols A1, A2, and A3 are used to indicate passivated Mo<sub>2</sub>N prepared according to the heating rate and NH<sub>3</sub> velocity shown in the table; Mo<sub>2</sub>N-B is nonpassivated Mo<sub>2</sub>N.

The MoO<sub>3</sub> starting material (Aldrich 99.5%) was pressed into pellets, lightly crushed, and sieved to produce 0.5- to 0.8-mm particles. Then, the MoO<sub>3</sub> particles were charged in a quartz reactor which was placed inside the furnace. The temperature was controlled by a programmable temperature controller and the reaction temperature was measured by a thermocouple which was inside the MoO<sub>3</sub> bed. NH<sub>3</sub> (air liquid 99.95%) was introduced after removing moisture using a moisture filter (Chrompack, gas clean filter) and its flow rate was measured using a bubble flow meter. Molar hourly space velocities of 20–50 mol NH<sub>3</sub>/mol MoO<sub>3</sub>/h were used. After the reaction was completed, the reactor was quickly cooled to room temperature in flowing NH<sub>3</sub>. The nonpassivated sample was directly characterized by TPD and TPR. For passivating samples a He–O<sub>2</sub> (1%) mixture was flowed into the reactor at room temperature overnight.

The molybdenum nitride samples which were used in TPD and TPR measurements are shown in Table 1. Mo<sub>2</sub>N-A1, Mo<sub>2</sub>N-A2, and Mo<sub>2</sub>N-A3 were passivated Mo<sub>2</sub>N samples treated by a He–O<sub>2</sub> mixture after the preparation. A1, A2, and A3 represent samples that were prepared by using various heating rates and NH<sub>3</sub> velocities as shown in Table 1. Mo<sub>2</sub>N-B was nonpassivated Mo<sub>2</sub>N. The BET surface area and pore diameter of Mo<sub>2</sub>N were influenced strongly by the synthesis conditions.

### 2.2. TPD and TPR Characterization

A molybdenum nitride sample (0.2 g), charged in the quartz reactor, was heated to 120°C in He and held at this temperature for 1 h. After cooling to room temperature, the sample was heated at 10°C/min to 1030°C in a flow of He (for TPD) or H<sub>2</sub> (for TPR). The flow rate of He and H<sub>2</sub> was 40 ml/min. The effluent stream was monitored for various species as a function of temperature by using an on-line QMS 311 mass spectrometer equipped with a capillary inlet heated by heating tape.

The reductivity behavior of molybdenum nitride was also studied by using a thermal conductivity detector and a mixture of N<sub>2</sub>–H<sub>2</sub> (5 v/v%) as reducing gas (TPR-TC). In order

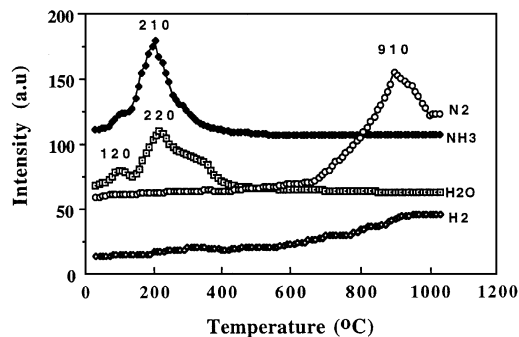


FIG. 1. TPD profile of the Mo<sub>2</sub>N-A1 sample (A1 refers to the preparation conditions shown in Table 1).

to measure H<sub>2</sub> consumption, a cooled trap on *n*-pentane-liquid N<sub>2</sub> (–127°C) was used to remove water formed in the reaction process and NH<sub>3</sub> desorbed from the samples.

## 3. RESULTS

### 3.1. TPD of Molybdenum Nitride

Figures 1–3 show the TPD profiles of passivated molybdenum nitride samples with different surface areas. Four curves of mass/e 2 (H<sub>2</sub>), 17 (NH<sub>3</sub>), 18 (H<sub>2</sub>O), and 28 (N<sub>2</sub>) vs temperature of TPD are reported. Obviously, the profiles of various species for Mo<sub>2</sub>N-A1, Mo<sub>2</sub>N-A2, and Mo<sub>2</sub>N-A3 are very similar except for peak intensities. The two peaks of H<sub>2</sub>O desorption appear at 120 and 220°C and the NH<sub>3</sub> desorption peak is around 220°C for all these samples. The peak intensities rise with increasing surface area of Mo<sub>2</sub>N. On the Mo<sub>2</sub>N-A3 sample, a small H<sub>2</sub> desorption peak at about 210°C can be also found, while no H<sub>2</sub> desorption peak appeared on low surface area molybdenum nitride. As for the profile of mass/e 28 (N<sub>2</sub>) vs temperature during TPD, the relative intensity of the N<sub>2</sub> signal began to increase gradually when the sample temperature was higher than 600°C. The maximum peak temperature of N<sub>2</sub> desorption is around 910°C. No influence of the surface area of Mo<sub>2</sub>N on the N<sub>2</sub> peak temperature is seen, but the peak intensities are much

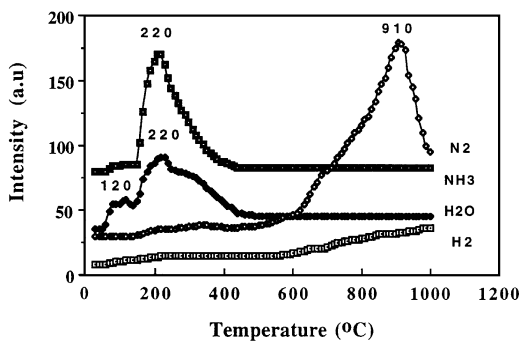


FIG. 2. TPD profile of the Mo<sub>2</sub>N-A2 sample (A2 refers to the preparation conditions shown in Table 1).

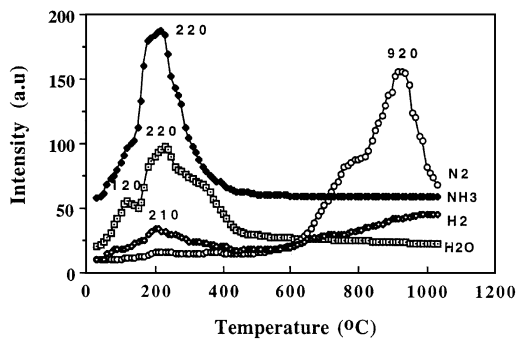


FIG. 3. TPD profile of the  $\text{Mo}_2\text{N-A3}$  sample (A3 refers to the preparation conditions shown in Table 1).

larger for the  $\text{Mo}_2\text{N-A2}$  and  $\text{Mo}_2\text{N-A3}$  samples which have relatively high surface areas. In addition, a small shoulder peak of  $\text{N}_2$  at about  $700^\circ\text{C}$  can be observed on  $\text{Mo}_2\text{N-A3}$ . The  $\text{N}_2$  peak appeared at so high a temperature that it is obviously not the result of the desorption of  $\text{N}_2$  adsorbed on molybdenum nitride. We suspect the  $\text{N}_2$  peak is from the  $\text{Mo}_2\text{N}$  decomposition at elevated temperatures.

In order to remove surface species, especially to reduce the surface oxygen formed during the passivating process, the passivated  $\text{Mo}_2\text{N-A2}$  sample was first pretreated at  $400^\circ\text{C}$  in  $\text{H}_2$  flow at 40 ml/min for 2 h and cooled to room temperature in  $\text{H}_2$ . The sample was then flushed with He for 4 h and heated in He flow at 40 ml/min from room temperature to  $1030^\circ\text{C}$  at  $10^\circ\text{C}/\text{min}$ . The results are shown in Fig. 4. The  $\text{NH}_3$  desorption peak has vanished, but two  $\text{H}_2$  desorption peaks appeared at 140 and  $480^\circ\text{C}$ . The former is a minor peak which is obviously due to the desorption of some  $\text{H}_2$  physically adsorbed or weakly chemisorbed on the  $\text{Mo}_2\text{N}$  surface. The latter with a large peak area could reasonably be attributed to the desorption of  $\text{H}_2$  strongly bonded to  $\text{Mo}_2\text{N}$ . The complete desorption of this  $\text{H}_2$  needs a much higher temperature, and even when the prerduced  $\text{Mo}_2\text{N}$  solid was first flushed with He at  $400^\circ\text{C}$  for 2 h, a  $\text{H}_2$  desorption peak is also observed during a following TPD process. Of course, the peak intensity is greatly decreased as shown in Fig. 5. It is suggested that the  $\text{H}_2$  which desorbed at

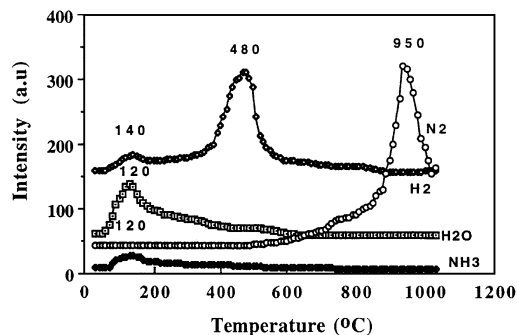


FIG. 4. TPD profile of the  $\text{Mo}_2\text{N-A2}$  sample prerduced at  $400^\circ\text{C}$  in  $\text{H}_2$  for 2 h.

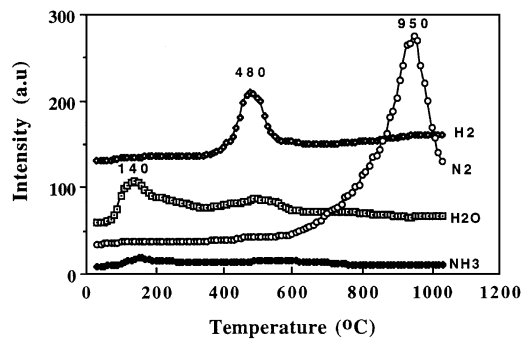


FIG. 5. TPD profile of the  $\text{Mo}_2\text{N-A2}$  sample following reduction at  $400^\circ\text{C}$  in  $\text{H}_2$  and flushed at  $400^\circ\text{C}$  by He for 2 h.

a high temperature is strongly bonded to the  $\text{Mo}_2\text{N}$  surface or in the crystalline phase of  $\text{Mo}_2\text{N}$  (15).

An obvious influence of the prerduction treatment on the  $\text{N}_2$  desorption peak is also seen in Figs. 4 and 5. Comparing these with the  $\text{Mo}_2\text{N}$  sample without  $\text{H}_2$  pretreatment, the maximum peak temperature of  $\text{N}_2$  over the  $\text{Mo}_2\text{N}$ -prerduced sample shifted from  $910$  to  $950^\circ\text{C}$ . It appears that the removal of surface oxygen on  $\text{Mo}_2\text{N}$  plays a role in inhibiting the decomposition of  $\text{Mo}_2\text{N}$ .

The TPD profile of fresh  $\text{Mo}_2\text{N}$  is shown in Fig. 6. A large difference can be observed as compared with Figs. 1–3. For the nonpassivated  $\text{Mo}_2\text{N}$  sample, the  $\text{NH}_3$  desorption peak is broad and around  $280^\circ\text{C}$ . A major  $\text{H}_2$  desorption peak at  $550^\circ\text{C}$  appeared. On the passivated  $\text{Mo}_2\text{N}$  sample the  $\text{H}_2$  desorption peak is hardly observed in the TPD process. The  $\text{N}_2$  spectra of  $\text{Mo}_2\text{N-B}$  exhibited much larger differences with  $\text{Mo}_2\text{N-A}$  series samples. Three  $\text{N}_2$  peaks at 560, 720, and  $960^\circ\text{C}$  were observed.  $\text{N}_2$  desorption at low temperatures was observed together with the  $\text{H}_2$  desorption peak. As  $\text{Mo}_2\text{N-B}$  was not passivated after the preparation in  $\text{NH}_3$  flow, it should be suggested that large quantities of  $\text{NH}_3$  remain adsorbed on the sample surface. Then, the  $\text{N}_2$  peak at  $550^\circ\text{C}$  could be considered a result of  $\text{NH}_3$  decomposition at an elevated temperature. Two other peaks would be associated with a change of the  $\text{Mo}_2\text{N}$  crystallite itself. The  $\text{N}_2$  peak at  $960^\circ\text{C}$  is obviously associated with the

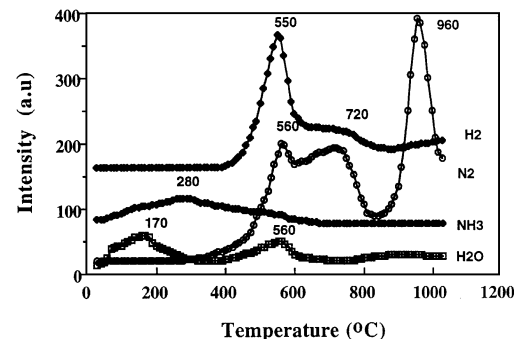


FIG. 6. TPD profile of the  $\text{Mo}_2\text{N-B}$  sample.

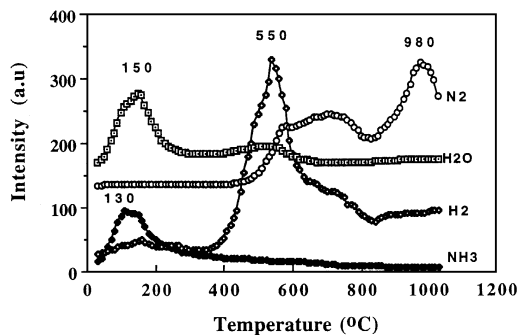


FIG. 7. TPD profiles of the Mo<sub>2</sub>N-B sample prerduced at 400°C in H<sub>2</sub> for 2 h.

decomposition of Mo<sub>2</sub>N and is similar to that of the passivated Mo<sub>2</sub>N sample. The broad peak of N<sub>2</sub> around 720°C demonstrated that some changes of the Mo<sub>2</sub>N crystallite phase occurred.

Figure 7 illustrates the TPD of the nonpassivated Mo<sub>2</sub>N sample prerduced at 400°C in H<sub>2</sub>. It is seen that a minor NH<sub>3</sub> desorption peak appeared below 200°C, and the H<sub>2</sub> desorption peak around 550°C became much larger as compared with Fig. 6. A large change in the N<sub>2</sub> desorption peaks was also observed. The intensity of the N<sub>2</sub> peak around 570°C decreased obviously. Since this N<sub>2</sub> peak was suggested as due to decomposition of NH<sub>3</sub>, a plausible explanation is that most of the NH<sub>3</sub> was desorbed during H<sub>2</sub> prerduction of Mo<sub>2</sub>N.

### 3.2. TPR of Molybdenum Nitride

The TPR-MS profiles for passivated Mo<sub>2</sub>N samples are shown in Fig. 8. The change in the H<sub>2</sub> signals was too small to be detected in H<sub>2</sub> flow, and only NH<sub>3</sub>, H<sub>2</sub>O, and N<sub>2</sub> spectra are observed. It is noticed that the N<sub>2</sub> signal in the temperature region 600–800°C increased slowly, but when the temperature was higher than 800°C, the N<sub>2</sub> signal quickly rose and exhibited a sharp peak at 870°C. This suggests that Mo<sub>2</sub>N would be rapidly decomposed in H<sub>2</sub> beyond a definite temperature. Compared to the profile of TPD, the temperature of NH<sub>3</sub> decomposition dropped about 40–50°C. In addition, a broadened NH<sub>3</sub> desorption peak appeared at 220°C.

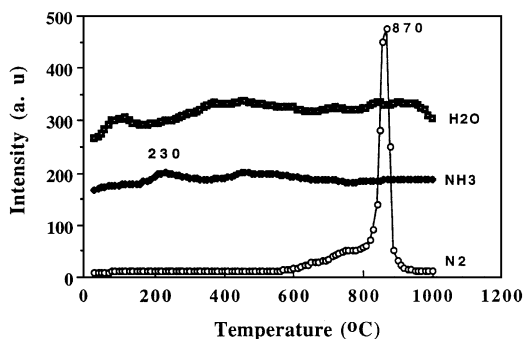


FIG. 8. TPR profile of the Mo<sub>2</sub>N-A2 sample.

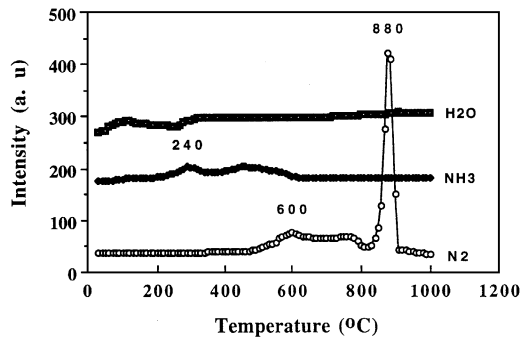


FIG. 9. TPR profile of the Mo<sub>2</sub>N-B sample.

The TPR-MS profile of the fresh Mo<sub>2</sub>N sample as shown in Fig. 9 is similar to that of the passivated Mo<sub>2</sub>N sample in Fig. 8. It means that no large difference exist between fresh and passivated Mo<sub>2</sub>N samples under H<sub>2</sub> reaction. Of course, for the fresh Mo<sub>2</sub>N sample a broad peak in the N<sub>2</sub> signal is more evident in the region 600–800°C. This means there exists the liberation of N<sub>2</sub> in a broad temperature region before the complete decomposition of Mo<sub>2</sub>N.

The crystallite changes in the nonpassivated Mo<sub>2</sub>N sample after TPD to 850 or 1030°C in He and TPR to 1000°C at 10°C/min were examined by XRD. The results are shown in Fig. 10. For the passivated Mo<sub>2</sub>N sample the characteristic

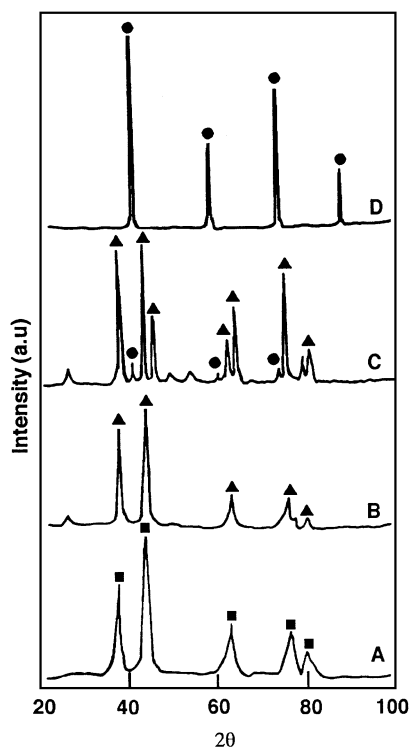


FIG. 10. XRD patterns of Mo<sub>2</sub>N samples. (A) Passivated Mo<sub>2</sub>N; (B) heated to 850°C in He; (C) heated to 1030°C in He; (D) heated to 1000°C in H<sub>2</sub>. ■, γ-Mo<sub>2</sub>N; ▲, β-Mo<sub>2</sub>N; ●, metallic Mo.

bands of  $\gamma$ - $\text{Mo}_2\text{N}$  are shown in Fig. 10A. When it was heated to  $850^\circ\text{C}$  in He,  $\gamma$ - $\text{Mo}_2\text{N}$  was transformed into  $\beta$ - $\text{Mo}_2\text{N}$  ( $\text{Mo}_{16}\text{N}_7$ ). When the temperature was increased to  $1030^\circ\text{C}$ , the resulting solid consisted of mixed  $\beta$ - $\text{Mo}_2\text{N}$  and metallic Mo crystallites. Reduction up to  $1000^\circ\text{C}$  reduced  $\text{Mo}_2\text{N}$  completely to metallic Mo. The results demonstrated that the changes in the crystalline phase of  $\text{Mo}_2\text{N}$  would occur before its decomposition at elevated temperatures.

The reaction behavior of the  $\text{Mo}_2\text{N}$  sample with  $\text{H}_2$ , using a  $\text{N}_2$ - $\text{H}_2$  (5%) mixture as the reducing gas and a thermal conductivity cell as the detector, was also examined. The TPR-TC profiles of  $\text{Mo}_2\text{N}$  samples are shown in Figs. 11 and 12. In this situation only  $\text{H}_2$  signal changes can be observed since all the other gases have been trapped. For the passivated  $\text{Mo}_2\text{N}$  samples with different surface areas, two temperature regions are evident as shown in Fig. 11, i.e., a low-temperature region around  $465^\circ\text{C}$  and a high-temperature region around  $840^\circ\text{C}$ . Obviously, the peak in the low-temperature region was due to the reduction of surface oxygen on  $\text{Mo}_2\text{N}$  formed during the passivating process. The peak in the high-temperature region is attributed to the results of the liberation of  $\text{N}_2$  due to  $\text{Mo}_2\text{N}$  decomposition. For the nonpassivated  $\text{Mo}_2\text{N}$  sample, three temperature regions are observed: (i) two  $\text{H}_2$  consumption peaks around  $300$  and  $500^\circ\text{C}$  appeared, (ii) a minor  $\text{H}_2$  evolution peak is seen in the region  $550$ – $700^\circ\text{C}$ , and (iii) a major  $\text{N}_2$  liberation peak appeared around  $840^\circ\text{C}$ , which is the same as observed on the passivated  $\text{Mo}_2\text{N}$  sample. The appearance of the  $\text{H}_2$  consumption peak in the low-temperature region revealed that some  $\text{H}_2$  consuming species are present on the nonpassivated  $\text{Mo}_2\text{N}$  sample. It is suggested that the

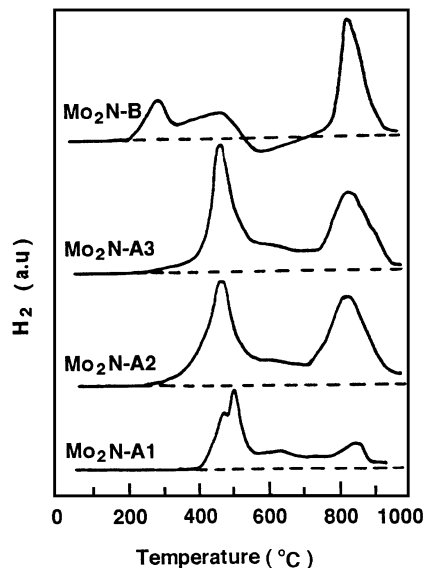


FIG. 11. TPR-TC profile of passivated  $\text{Mo}_2\text{N}$  (A1, A2, A3) and fresh  $\text{Mo}_2\text{N}$  (B).

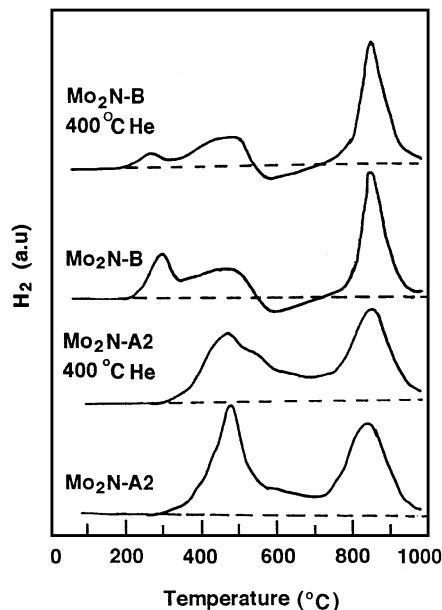


FIG. 12. The influences of pretreatment at  $400^\circ\text{C}$  in He on TPR-TC profiles of  $\text{Mo}_2\text{N}$ .

desorption of  $\text{NH}_x$  species ( $x = 1$ – $3$ ) adsorbed on the  $\text{Mo}_2\text{N}$  surface and the reduction of a residual trace of oxide during  $\text{MoO}_3$  nitridation are probably responsible for the  $\text{H}_2$  consumption. The broad  $\text{H}_2$  evolution peak is due to either the desorption of  $\text{H}_2$  adsorbed on the  $\text{Mo}_2\text{N}$  sample or the release of  $\text{H}_2$  upon  $\text{NH}_3$  decomposition, which is observed only on the nonpassivated  $\text{Mo}_2\text{N}$  sample. This is consistent with the TPD-MS data.

Figure 12 presents the results of first pretreating the  $\text{Mo}_2\text{N}$  sample at  $400^\circ\text{C}$  in He flow for 2 h, and then examining it by TPR-TC. For the passivated  $\text{Mo}_2\text{N}$  sample, the  $\text{H}_2$  consumption peak of the low-temperature region became slightly broadened, while for the nonpassivated  $\text{Mo}_2\text{N}$  sample, the intensity of the  $\text{H}_2$  peak at  $300^\circ\text{C}$  decreased greatly.

#### 4. DISCUSSION

##### 4.1. Surface Species on Molybdenum Nitride

Due to the high surface area of  $\text{Mo}_2\text{N}$  and the method of preparation by reaction of  $\text{MoO}_3$  and  $\text{NH}_3$ , it is reasonable to consider that ammonia adsorbed on the  $\text{Mo}_2\text{N}$  surface. It was reported that a considerable amount of  $\text{NH}_3$  is desorbed from passivated  $\text{Mo}_2\text{N}$  samples in the temperature region  $200$ – $300^\circ\text{C}$  (16). Nagai *et al.* also observed the presence of  $\text{NH}_3$  desorption peaks on nitrated molybdena-alumina (12). Haddix *et al.* suggested that ammonia can be held in several states, such as  $\text{NH}_3$ ,  $\text{NH}_2$ , and  $\text{NH}$ , on the  $\text{Mo}_2\text{N}$  surface (15). Thus, it is suggested that  $\text{NH}_x$  species always remain on the  $\text{Mo}_2\text{N}$  surface. From our results, the differences in  $\text{NH}_x$  species adsorbed on the passivated  $\text{Mo}_2\text{N}$  and on

the nonpassivated Mo<sub>2</sub>N samples are evident. For Mo<sub>2</sub>N-passivated samples, only a NH<sub>3</sub> desorption peak around 220°C was observed as shown in Figs. 1–3. When the temperature was beyond 400°C, the NH<sub>3</sub> desorption curve became a horizontal line. However, for the nonpassivated Mo<sub>2</sub>N sample, the ammonia desorption peak is very broad (from 100 to 600°C). We suspect that another NH<sub>x</sub> species, which would decompose to N<sub>2</sub> and H<sub>2</sub> at elevated temperatures, exists. In Fig. 6, the N<sub>2</sub> and H<sub>2</sub> desorption peaks around 560°C actually come from the decomposition of NH<sub>x</sub> adsorbed on the sample surface. Therefore, it is suggested that two kinds of NH<sub>x</sub> species remain on the nonpassivated Mo<sub>2</sub>N surface. The first kind, desorbed below 300°C, is weakly chemisorbed. Another kind, strongly chemisorbed on the surface, was formed during the synthesis of Mo<sub>2</sub>N at 700°C under NH<sub>3</sub>. For the passivated Mo<sub>2</sub>N, only a weakly adsorbed ammonia species is present on the surface.

Studies have shown that molybdenum nitride exhibits significant capacities for adsorbing hydrogen (17). It was suggested that irreversibly held H<sub>2</sub>, strongly bonded to the nitride surface, may occur on nitrogen-deficient patches of Mo present on the nitride surface. For the passivated Mo<sub>2</sub>N sample a small peak of weakly adsorbed H<sub>2</sub> was observed only on the high surface area sample. It means that no significant amount of H<sub>2</sub> remained on the passivated Mo<sub>2</sub>N. When this Mo<sub>2</sub>N sample was reduced in H<sub>2</sub> at 400°C, two H<sub>2</sub> desorption peaks were evident (Fig. 4). The minor H<sub>2</sub> peak desorbed at a low temperature could be considered as adsorbed on the surface of Mo nitride, while H<sub>2</sub> desorbed at a high temperature could probably be attributed to unsaturated coordinative sites on the surface, or in the subsurface sides or bulk of Mo<sub>2</sub>N. The second type of hydrogen can be retained up to 480°C. In TPD profiles of freshly prepared Mo<sub>2</sub>N, except for H<sub>2</sub> release around 550°C, there is no other H<sub>2</sub> desorption peak. We suspect that, as the freshly prepared Mo<sub>2</sub>N was cooled to room temperature in NH<sub>3</sub> flow, most of the H<sub>2</sub> formed during the synthesis of Mo<sub>2</sub>N at high temperatures could be flushed out by NH<sub>3</sub> in the decreasing temperature process. Another reason is that a part of H<sub>2</sub> adsorbed on Mo<sub>2</sub>N could react with NH<sub>2</sub> or NH species existing on the Mo<sub>2</sub>N surface forming NH<sub>3</sub>. Comparing Figs. 6 and 7, it is seen that when the fresh Mo<sub>2</sub>N was reduced in H<sub>2</sub> at 400°C, the area of the H<sub>2</sub> desorption peak around 550°C is much larger. The increase of H<sub>2</sub> is attributed to the desorption of H<sub>2</sub> remaining on Mo<sub>2</sub>N during reduction.

In conclusion, there are two types of NH<sub>x</sub> adsorbed species that are weakly adsorbed and strongly bonded on the fresh Mo<sub>2</sub>N surface. The latter can be decomposed to N<sub>2</sub> and H<sub>2</sub> around 560°C. A kind of H species probably occupies nitrogen-deficient patches of Mo on the surface or enter into the lattice of nitride. The passivated Mo<sub>2</sub>N solid has only a weakly adsorbed NH<sub>x</sub> species, which would desorb below 300°C.

#### 4.2. The Influence of Passivation Treatment on Mo<sub>2</sub>N

Some unsaturated coordinative sites remain on the fresh Mo nitride surface. The NH<sub>x</sub> or H species formed during NH<sub>3</sub> decomposition at elevated temperatures could be strongly adsorbed on these sites. A considerable amount of active H atoms can also enter into the lattice of Mo nitride (17). A rapid autothermal oxidation of adsorbed NH<sub>x</sub> species and hydrogen species would result upon exposure to air.

Furthermore, oxygen incorporated during passivation forms a protective layer on the surface. It has been suggested that (14, 15, 18, 19) passivated oxygen is chemisorbed on the nitride surface, and one or two monolayers of oxide are formed, or oxygen diffuses into the lattice forming an oxynitride. Our TPR-TC data indicated that the reduction peak of the oxide forming the passivation layer appeared around 470°C, which is almost the same as the reduction temperature of the M=O bond in MoO<sub>3</sub> (20). This suggests that the oxygen was bonded to Mo atoms on top sites, which was considered to be where NH<sub>x</sub> was bonded originally (15). Of course, it is not excluded that a part of the surface layer oxygen diffuses into the subsurface of Mo nitride, since the TPR reduction peak is very wide.

The reduction treatment on Mo<sub>2</sub>N samples at elevated temperature removed the surface layer oxygen formed during passivation. It was observed in Fig. 4 that the H<sub>2</sub>O peak at 220°C was eliminated by oxygen removal. But the surface properties are modified after the removal of oxygen. The release of the N<sub>2</sub> peak was delayed to 950°C, and a large H<sub>2</sub> desorption peak appeared at 480°C. As for fresh Mo<sub>2</sub>N reduced in H<sub>2</sub> at 400°C, the H<sub>2</sub> desorption peak around 550°C became much larger, indicating that a large amount of hydrogen is retained in Mo<sub>2</sub>N.

We suspect that the main effect of passivation on freshly prepared Mo<sub>2</sub>N is to eliminate most of the NH<sub>x</sub> and H species strongly adsorbed on the unsaturated coordinative sites of nitride and to form a protective oxide layer. This oxygen can be reduced by hydrogen at elevated temperature, but the surface state of Mo<sub>2</sub>N reduced in H<sub>2</sub> at elevated temperatures is modified by the reduction.

### 5. CONCLUSION

The surface of fresh Mo nitride prepared by the temperature-programmed reaction of MoO<sub>3</sub> with NH<sub>3</sub> retains significant amounts of NH<sub>x</sub> and H species formed during NH<sub>3</sub> decomposition at elevated temperatures. The part that is weakly chemisorbed NH<sub>x</sub> species can be desorbed at temperatures lower than 300°C, while that part bonded strongly to Mo atoms is decomposed and forms H<sub>2</sub> and N<sub>2</sub> at temperatures over 550°C. When Mo nitride was passivated in a He–O<sub>2</sub> (1%) mixture, only a small amount of weakly adsorbed NH<sub>x</sub> was retained. The oxygen would be bonded

to Mo atoms to occupy the original positions of  $\text{NH}_x$  species and formed a protective layer. Part of the oxygen can diffuse into the subsurface of  $\text{Mo}_2\text{N}$ , but no separated  $\text{MoO}_3$  phase appeared. A major reduction peak of passivated Mo nitride was observed at  $470^\circ\text{C}$ . It means that the oxygen incorporated during passivation can be removed by  $\text{H}_2$  at moderate temperatures. The surface of reduced  $\text{Mo}_2\text{N}$  is modified by the reduction in  $\text{H}_2$ .

When fresh  $\text{Mo}_2\text{N}$  was heated in He over  $1000^\circ\text{C}$ , it would transform first from  $\gamma\text{-Mo}_2\text{N}$  to  $\beta\text{-Mo}_2\text{N}$ , and then gradually decompose to metallic Mo. For passivated  $\text{Mo}_2\text{N}$  no obvious changes in the crystalline phase occur before its decomposition.  $\text{Mo}_2\text{N}$  can be completely reduced to metallic Mo by hydrogen below  $1000^\circ\text{C}$ .

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